

The Bearing Axial Cracks Root Cause Hypothesis of



Frictional Surface Crack Initiation and Corrosion Fatigue Driven Crack Growth

Presented at the NREL 2011 Wind Turbine Tribology Seminar, Broomfield/CO Presented by <u>Jürgen Gegner</u> and Wolfgang Nierlich

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- 1. Appearance and Analysis of WEC Premature Bearing Failures
- 2. Literature Survey on Root Cause Hypotheses
- 3. Surface Cleavage Cracking & Corrosion Fatigue WEC Growth
 Ÿ Near-Surface Inhomogeneities as Crack Nuclei
 Ÿ No Distinct Crack Nuclei Present
- 4. Cleavage Crack Initiation by Frictional Tensile Stresses
 Ÿ Vibration Loading and Micro Friction Model
 Ÿ Normal Stress Hypothesis
- 5. Failure Prevention
- 6. Conclusions



Appearance and Analysis of WEC Premature Bearing Failures



axial raceway cracks of <1 mm to >20 mm length





axial raceway cracks of <1 mm to >20 mm length**Y** partly with shell-shaped spallings







axial raceway cracks of <1 mm to >20 mm length Ÿ partly with shell-shaped spallings from crack returns







axial raceway cracks of <1 mm to >20 mm length Ÿ partly with shell-shaped spallings from crack returns







axial raceway cracks of <1 mm to >20 mm length \ddot{Y} partly with shell-shaped spallings \ddot{Y} ... to advanced spallings







White etching cracks (WEC) bearing failures occur ...

- Ÿ at 1% to 20% of L_{10}^{nom} , i.e. before the *failure-free* time ú premature bearing failures ⇒ not ordinary RCF as root cause
- İn industrial gearboxes, cranes, paper making machines, dryers, ship drives, mill drives, coal pulverizers, generators, ...
 ú not restricted to wind turbines
- Ÿ by trend increasingly with enhanced power of the wind turbineú solution most important for offshore wind energy generation
- Ÿ rarely but in all wind turbine gearbox locations
- Ÿ basically independent of heat treatment and bearing type



case hardening – CARB Ÿ tends to spalling rather than through cracking



Ÿ basically independent of heat treatment and bearing type



bainite hardening – SRB Ÿ tends to spalling rather than through cracking



Ÿ basically independent of heat treatment and bearing type



bainite hardening – CRBŸ development of the crack system on the raceway





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bainite hardening – CRB
Ÿ development of the crack system on the raceway
ú crack returns indicated





Ÿ basically independent of heat treatment and bearing type



bainite hardening – CRB

- **Ÿ** strongly branching & spreading deep transgranular crack systems at moderate permanent loads
 - **ú** chemically assisted RCF \Rightarrow corrosion fatigue cracking (CFC)
- $\ddot{\mathbf{Y}}$... in overrolling direction \Rightarrow surface initiation indicated



Ÿ basically independent of heat treatment and bearing type





Y basically independent of heat treatment and bearing type



Failure Analysis

methods to study material loading and WEC damage mechanisms

- **Ÿ** visual and SEM raceway inspection
 - ú surface condition, tracing of crack paths, crack detection
- ÿ preparative crack opening
 ú fractographic SEM investigation and microchemical analysis
- W metallography (incl. SEM): axial & <u>circumferential</u> microsections
 ú characterization of crack systems and microstructural changes
- Ÿ spatially resolved determination of the hydrogen content
- Ÿ X-ray diffraction residual stress material response analysis
 ú clarification of the loading conditions



WEC Root Cause Hypotheses – A Commented Literature Survey



WEC Root Cause Hypotheses

prevailing opinions in the literature

- Ÿ subsurface failure due to
 - ú continuous hydrogen absorption through the rolling contact
 - ú abnormal butterfly growth
- **Ÿ** adiabatic shear band formation



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subsurface cracks tend to develop large fatigue spallings when reaching the raceway surface **Ÿ** extended spalling-free crack systems not to be expected





virtually undetectable hairline cracks



local surface smoothing reveals raceway hairline crack in the SEM



 virtually undetectable hairline cracks can cause large WEC systems
 Y negative way of conclusion from surface connection not found to not there is inconsistent due to limited detection probability





OD shows surface initiation – but no raceway crack visible



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- **Ÿ** *negative* way of conclusion from surface connection not found to not there is inconsistent due to limited detection probability
- Y positive way of conclusion from surface initiation verified via CFC to resulting crack-WEA arrangements is more comprehensible





OD shows surface initiation – crack opened during preparation



virtually undetectable hairline cracks can cause large WEC systems

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e.g. spidery WEA patterns: crack growth prior to phase change



hydrogen absorption unlikely through an undamaged RC surface **Ÿ** effect known from HF current passage or dense raceway cracks **ú** HT (case hardening)



raceway in the contact zone OR-DGBB from alternator rig

> 2011-11-16 © SKF Group Material Physics Schweinfurt

CGHE: $c_{\rm H} > 3^{\pm 0.2}$ ppm



hydrogen absorption unlikely through an undamaged RC surface
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 ú H-RCF: SWB @ b/B≈0.71, microcracking



OR-DGBB from alternator rig



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 $\mathbf{\hat{u}}$ accelerated increase of dislocation density and glide mobility \Rightarrow DGSL mechanism & HELP



dislocation-carbide interaction





hydrogen absorption occurs rapidly not until raceway cracking



Ÿ D_{eff} ≈ 10⁻⁷ cm²/s ⇒ just several weeks to few months of inward diffusion after crack initiation and growth

- Ÿ hydrogen released from aging products of penetrating lubricant
- **Ÿ** intergranular hydrogen embrittlement of opened fracture faces is restricted to the forced rupture around original cracks



butterfly formation is included in the bearing life theory $\ddot{\mathbf{Y}}$ rapid generation at $p_0 \ge 1400$ MPa but slow growth $\acute{\mathbf{u}}$ distribution follows the (alternating) orthogonal shear stress $\ddot{\mathbf{Y}}$ the rare failures occur in the upper life range above \mathcal{L}_{10}^{nom}



WEC occur with or without butterflies: $p_0 > / < 1400$ MPa \ddot{Y} evolution of butterfly population up to WEC must be traceable



evaluation of 101 butterflies ú exponential PDF describes natural growth



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WEC occur with or without butterflies: $p_0 > / < 1400$ MPa \ddot{Y} evolution of butterfly population up to WEC must be traceable butterflies reveal no DER precursor structure of WEA formation \ddot{Y} ... but WEA of WEC do (hydrogen from aging oil in CFC cracks)



<u>50 μm</u>



WEC occur with or without butterflies: $p_0 > /< 1400$ MPa \Rightarrow WEC and butterflies are independent microstructural changes butterflies reveal no DER precursor structure of WEA formation







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WEC Root Cause Hypotheses – Adiabatic Shearing

local flash austenitizing by very rapid large plastic deformation

- Ÿ shock straining conditions do not occur in bearing operation
- Ÿ real bearing dynamics do not create critical EHL pressure peaks
 ú impact loads act like small Hertzian contacts or indentations
 ú w/o frictional surface traction, no tensile stress occurs



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- **Ÿ** ASB represent essentially straight regular ribbons of mm length





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Note: adiabatic shearing also presumed to cause the regular SWB Ÿ gradual development of SWB rather indicates RCF damage Schlicht, 2008



Spontaneous Axial Surface Cracking and Corrosion Fatigue WEC Growth





Typical raceway cracks to be opened for fractography

. . .



small crack < 1mm



large cracks > 20 mm



Small raceway crack



• original fracture surface clearly distinguishable



Small raceway crack



- original fracture surface clearly distinguishable
- trapezoid shape of the spreading crack reveals top-down growth





Small raceway crack



- original fracture surface clearly distinguishable
- trapezoid shape of the spreading crack reveals top-down growth





Small raceway crack



... further crack growth by corrosion fatigue



Small raceway crack

Large raceway crack







Small raceway crack

Large raceway crack



• radial propagation lines \Rightarrow top-down growth, surface initiation



Description of the cracks



- step 1: cleavage-like fractures → normal stress hypothesis
- step 2: corrosion fatigue cracking (CFC)
- \rightarrow microfractography, phase changes





inhomogeneity: small butterfly on MnS inclusion

cleavage fracture as initiation of CFC



Vega ©Tescan

SKF



View field: 113.34 um DET: SE Detector HV: 20.0 kV

50 um

partly radial fan structure

to the right of the butterfly: CFC









inhomogeneity: inclusion TiCN + MnS

cleavage fracture as initiation of CFC





inhomogeneity: inclusion TiCN + MnS

circular low-deformation vertical lenticular incipient crack below an axial raceway crack

cleavage fracture as initiation of CFC





inhomogeneity: inclusion TiCN + MnS

edge-tracing bulge verifies that lens crack occurs first

cleavage fracture as initiation of CFC





inhomogeneity: inclusion TiCN + MnS

cleavage fracture as initiation of CFC





inhomogeneity: inclusion TiCN + MnS

cleavage fracture as initiation of CFC



distinct change of the fracture pattern also on the left of the lens



fan structure on *cleavage* area

faint side cracks outside







inclusion, no butterfly

start of fan structure

detail 2







inhomogeneity: near-surface tensile residual stress

cleavage fracture as initiation of CFC





Vega ©Tescan

SKF



View field: 113.13 um DET: SE Detec HV: 20.0 kV

50 um

faint radial fan structure

outside of *cleavage*: side cracks, martensite microstructure partly visible

2011-11-16 © SKF Group

Material Physics Schweinfurt

detail





axial raceway crack

 $\ddot{\mathbf{Y}}$ distinctly smoothed honing structure \Rightarrow mixed friction conditions

Ÿ brittle spontaneous incipient crack below sharp-edged segment
 ú clearly distinguishable from adjacent serrated branching CFC



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overview of the opened crack

- Ϋ́ crack depth about 900 μm
- **Ÿ** preparative forced fracture clearly distinguishable



1 mm Vega ©Tescan SKF

... brittle incipient *cleavage* crack





different crack mechanisms inside and outside the flat *cleavage* fracture surface Ÿ 130 µm deep brittle vertical incipient crack Ÿ surrounded by striation-like lines of CFC



SEM-SE









preparatively generated forced fracture face reveals local hydrogen embrittlement adjacent to the original CFC crack







preparatively generated forced fracture face reveals local hydrogen embrittlement adjacent to the original CFC crack Ÿ increased intergranular fraction







dense surface cracks: H absorption





- **Ÿ** S1 at incipient *cleavage* crack: only subsequent surface corrosion
- **Ÿ** S2-S6 reveal much stronger signals of the typical tracer elements sulfur, phosphor and zinc of the lubricant additives
 - ú higher tracer enrichment in the (deeper) CFC region due to mixed mechanical-chemical loading





cleavage fracture as initiation of CFC

different crack mechanisms inside and outside the flat face





demarcating crack network at the runout of the *cleavage* fracture

detail 1

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demarcating crack network
Ÿ *cleavage* crack occurs first
Ÿ temporary crack stop and gradually starting CFC

detail 2






origin of the crack network demarcating the cleavage fracture

 embrittled DER precursor of WEA evolution on CFC cracks ú preparative etching cracks reveal material embrittlement (SEM)







origin of the crack network demarcating the *cleavage* fracture

hydrogen induced microstructure transformation

 hydrogen released from decomposition products of the
 penetrating oil (additives, contaminations) on the rubbing
 blank metal crack faces/tip





H induced DER: deformation bands as starting crack network

- occasionally visible as topographic feature at fracture faces
- carbide dissolution (DER precursor → WEA by DRX)
 ú localized H-RCF (DGSL & HELP)











origin of hydrogen sorption







origin of hydrogen sorption

local H embrittlement

30 mm mean ring thickness` CGHE: ± 0.2 ppm H 3 mm sample height

Raceway	1.8 ppm H	Raceway
Center	1.4 ppm H	Center
Bore	0.6 ppm H	Bore





origin of hydrogen sorption

local H embrittlement

is dense *cleavage* surface cracking (penetrating lubricant on CFC cracks)

30 mm mean ring thickness CGHE: ± 0.2 ppm H 3 mm sample height

Raceway	1.8 ppm H	Raceway
Center	1.4 ppm H	Center
Bore	0.6 ppm H	Bore

 ⇒ just several weeks to to few months of hydrogen in-diffusion







vertical semicircular *cleavage* fracture

• sharp crack edges indicate the incipient hairline *cleavage*









• side cracks and pores at the runout of the *cleavage* fracture







2011-11-16 © SKF Group Material Physics Schweinfurt step 1

step 2

total fracture face is transgranular



EPMA-EDX (10 kV)



600 µm

lens and CFC deep crack











S1

Zn Zn







4 *Cleavage* Crack Initiation by Frictional Tensile Stresses



Crack Initiation by Frictional Tensile Stresses 1. CRS Buildup near Indentation-Free Raceway Surfaces

material loading by equivalent shear stresses caused by vibrations
 ÿ both types of vibration residual stress pattern occur in WEC cases
 ú J. Gegner and W. Nierlich: AXA 52 (2008) 722-731



Crack Initiation by Frictional Tensile Stresses 2. Equivalent Stress Distribution in Rolling-Sliding Contact

Broszeit et al., 197

material loading by equivalent shear stresses caused by vibrations
 ÿ both types of vibration residual stress pattern occur in WEC cases
 ú tribological model of localized friction coefficient



increasing friction shifts the max. equivalent stress to the surface

a semiminor axis of the contact area



Crack Initiation by Frictional Tensile Stresses 2. Equivalent Stress Distribution in Rolling-Sliding Contact

material loading by equivalent shear stresses caused by vibrations
 ÿ both types of vibration residual stress pattern occur in WEC cases
 ú tribological model of localized friction coefficient



increasing friction shifts the max. equivalent stress to the surface $\Rightarrow \mu_{>} \ge 0.4$ occurs

- intermittently varying $\mu_{>}$ and $\mu_{<}$ $\Rightarrow \mu_{(eff)} < 0.1$
- shear sensitive viscosity

Broszeit et al., 1977



Crack Initiation by Frictional Tensile Stresses 2. Equivalent Stress Distribution in Rolling-Sliding Contact

material loading by equivalent shear stresses caused by vibrations
 ÿ both types of vibration residual stress pattern occur in WEC cases
 ú tribological model of localized friction coefficient



type B
$$\xrightarrow{\mu_> > 0.25}$$
 type A

- intermittently varying $\mu_{>}$ and $\mu_{<}$ $\Rightarrow \mu_{(eff)} < 0.1$
- shear sensitive viscosity

Broszeit et al., 1977



Crack Initiation by Frictional Tensile Stresses 3. Tangential Tensile Stresses at the Contact Runout





2011-11-16 © SKF Group Material Physics Schweinfurt Karas, 1941





IR-TRB, wind turbine gearbox







IR-TRB, wind turbine gearbox







IR-TRB, wind turbine gearbox









WEC Early Failure Prevention – Measures against Surface Cracks



compressive residual stresses by cold working Ÿ successfully proven in mixed friction loaded rolling contact





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roller-on-disk rig test

compressive residual stresses by cold working Ÿ successfully proven in mixed friction loaded rolling contact ú cleavage cracks reproduced





roller-on-disk rig test



compressive residual stresses by cold working Ÿ successfully proven in mixed friction loaded rolling contact ú cleavage cracks reproduced





roller-on-disk rig test



compressive residual stresses by cold working

- **Ÿ** successfully proven in mixed friction loaded rolling contact
 - ú cleavage cracks reproduced and prevented by deep rolling



compressive residual stresses by cold working

- **Ÿ** successfully proven in mixed friction loaded rolling contact
 - ú cleavage cracks reproduced and prevented by deep rolling
- + black oxidizing: reduced micro friction under peak loading
- + thermal post-treatment





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- **Ÿ** successfully proven in mixed friction loaded rolling contact
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 - **ú** reheating below HT tempering/transformation temperature
 - ú microstructure stabilization by thermal static strain aging
 - ú after grinding reported as full solution against WEC (Luyckx, 2011)

compressive residual stresses by cold working

- **Ÿ** successfully proven in mixed friction loaded rolling contact
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- + black oxidizing: reduced micro friction under peak loading
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promising concept for WEC resistant wind turbine gearbox bearings









Conclusions

- Ÿ results of failure analysis and research are presented
- Ÿ axial raceway cracks in some medium and large size bearings
- Ÿ root cause hypotheses from the literature are reviewed
- **Ÿ** crack initiation by *cleavage* fracture and propagation by CFC
- **Ÿ** hydrogen impact due to lubricant reactions at CFC crack tip/faces
- Y material response in the form of *cleavage* cracking suggests tangential tensile stresses acting, e.g., on weaker strength areas ú inhomogeneities, near-surface material aging (embrittlement)
- **Ÿ** tensile stresses are estimated to be high enough for cracking
- **Ÿ** tensile stresses are caused by sliding friction e.g. due to vibrations
- **Ÿ** cold working CRS & black oxidizing & reheating is proposed as effective countermeasure


Conclusions

- Ϋ a textbook chapter on bearing failures & rolling contact tribology is available for free download:
 - **Ú** <u>http://www.intechopen.com/articles/show/title/tribological-aspects-of-rolling-bearing-failures</u>



